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Valleytronics is a particularly interesting field that employs the valley degree of freedom for information manipulation. The fascinating prospects for realizing valleytronic devices have inspired persistent efforts towards exploring material systems with robust valley polarization. Monolayer transition metal dichalcogenides (TMDs) obey the well-known valley-dependent selection rule as a result of their inversion asymmetry. However, for inversionsymmetric bilayer tungsten-based TMDs, highly selective valley polarization has been surprisingly observed and is not yet fully understood. Here we systematically study the origin of the anomalously high valley polarization in bilayer WS₂ by temperaturedependent polarization-resolved photoluminescence measurements. It is found that acoustic phonons play a critical role in the valley polarization of bilayer WS₂. For some WS₂ bilayers with relatively small intensity ratios of indirect to direct bandgap emission, acoustic phonons could remarkably assist the intervalley scattering process and smear the valley contrast. On the other hand, in other bilayers, which show obvious indirect band gap emission, the indirect optical transition process depletes the phonon mode at the Λ point dramatically and results in anomalously robust valley polarization in bilayer WS₂. These results help recognize the crucial role of electronphonon coupling in intervalley relaxation in bilayer WS₂ and provide new insights into the future design of valleytronic devices based on two-dimensional TMDs.

Unveiling exceptionally robust valley contrast in AA- and AB-stacked bilayer WS₂†

Yanlong Wang, ^(b) ^{ab} Chunxiao Cong,*^c Jingzhi Shang,^a Mustafa Eginligil,^d Yuqi Jin,^b Gang Li,^b Yu Chen,^a Namphung Peimyoo^a and Ting Yu ^(b)*^a

Conceptual insights

Valleytronics has emerged as a particularly interesting field and shows fascinating prospects for realizing valleytronic devices. Monolayer transition metal dichalcogenides (TMDs) possess valley polarization as a result of their inversion asymmetry, which makes them promising for nextgeneration valleytronics. For bilayer TMDs with tunable interlayer interaction, valley polarization is more desirable and experimentally observed to be even larger than the monolayer counterpart in WS2 with inversion symmetry. However, the exact mechanism of the robust valley polarization in bilayers remains unknown, which greatly limits the development of valleytronic devices based on TMDs. We, for the first time, report the correlation between the intensity of the indirect bandgap emission and valley polarization. We have demonstrated that acoustic phonons play a leading role in the valley polarization of bilayer WS2 and the depletion of such phonons in the indirect optical transition process could contribute to the exceptionally robust valley polarization. The observed tunable valley polarization by interlayer coupling not only sheds light on the critical role of electron-phonon coupling in the intervalley scattering process in bilayer WS2, but also provides guidance for enhancing the valley contrast in few-layer TMDs, both of which are essential for the future development of novel valleytronic devices based on two-dimensional materials.

Introduction

Two-dimensional (2D) group-VIB transition metal dichalcogenides (TMDs) have generated considerable research interest in recent years for both fundamental studies and application prospects as a result of their inherent energy gaps and numerous other promising features such as the transformation from the indirect to direct bandgap when approaching the monolayer limit.^{1–9} In particular, semiconducting TMDs with the general formula of MX₂ (M represents Mo or W and X represents S or Se)^{6,10} are considered interesting candidates for achieving valleytronics,^{11,12} a rather interesting field where the valley degree of freedom is explored as an information carrier.¹³ In monolayer TMDs, there exist inequivalent and energetically degenerate valleys at *K* and *K'* points in the hexagonal Brillouin zone.^{14–16} It has been predicted that interband transitions obey the valley-dependent optical

^a Division of Physics and Applied Physics, School of Physical and Mathematical Sciences, Nanyang Technological University, 637371, Singapore. E-mail: yuting@ntu.edu.sg

^b Key Laboratory of Chemical Lasers, Dalian Institute of Chemical Physics, Chinese Academy of Sciences, Dalian 116023, China

^c State Key Laboratory of ASIC & System, School of Information Science and Technology, Fudan University, Shanghai 200433, China.

E-mail: cxcong@fudan.edu.cn

^d Key Laboratory of Flexible Electronics (KLOFE) & Institute of Advanced Materials (IAM), Jiangsu National Synergetic Innovation Centre for Advanced Materials (SICAM), Nanjing Tech University (NanjingTech), 30 South Puzhu Road, Nanjing 211816, China

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selection rule in non-centrosymmetric monolayer TMDs.^{17–20} More specifically, the direct bandgap transitions at *K* (*K'*) exclusively couple to σ_+ (σ_-) circularly polarized light.^{6,10} This valley-contrasted circular polarization, together with the large separation between *K* and *K'* points in the momentum space, makes the valley serve as another robust index of charge carriers in addition to spin.²⁰ Valley polarization could be detected by measuring the circular states of photoluminescence (PL) after selective optical pumping and the expected circular dichroism has been observed in monolayer TMDs in this way.^{14–16,21,22}

In bilayer TMDs with inversion symmetry, the valley-dependent selection rule does not apply anymore and low valley contrast was observed in AB-stacked bilayer MoS₂,^{15,23,24} which greatly restricts the range of thickness selection for achieving potential MoS₂-based valleytronics. In contrast to MoS₂ with comparable spin-valley coupling and interlayer hopping energy, the significantly higher spin-valley coupling strength in bilayer tungstenbased TMDs (WX₂) leads to suppressed interlayer hopping. Therefore, WX₂ bilayers possess an extra index called layer polarization in addition to valley and spin indices, which could provide information on the location of charge carriers (either top-layer or bottom-layer).14,25 The giant spin-layer-valley coupling paves a new way to manipulation of the electronic degrees of freedom in WX₂.^{14,25} Excitingly and surprisingly, highly selective circular polarization has been reported for the direct transition in centrosymmetric bilayer WS2 and WSe2,14,26,27 which is not only contrary to the widely accepted anticipation that valley-induced circular polarization should be inherently absent in centrosymmetric materials, but also shows an even larger valley contrast value than that of the monolayer,^{14,26} indicating that bilayer WX₂ is a fascinating platform for further exploration of valley physics.¹⁸ However, the unexceptionally large valley contrast in bilayer WX₂ is far from being well understood and deserves further investigation. The anomalous valley property was attributed to either localization of spin polarization in each individual layer by Liu et al.²⁸ or suppressed interlayer hopping as a result of large spin-valley coupling compared to the interlayer hopping amplitude by Zhu et al. in bilayer WX2.14,25 Though these explanations provide valuable information that valley polarization is not only limited to noncentrosymmetric systems in WX₂, why the valley polarization in bilayer WX2 could be obviously larger than that in the monolayer counterpart^{14,27} still cannot be explained from either of these two viewpoints and remains to be elucidated.

Interlayer coupling has been shown to significantly tune the optical properties of bilayer WX₂.^{29,30} However, up to now, the experimental studies of valley properties have been limited to mechanically exfoliated (ME) bilayer WS₂ samples with quite strong interlayer coupling.^{14,26,27} Though the interlayer coupling is expected to have a pronounced effect on valley polarization,²⁴ systematic studies of their relationship are still lacking in bilayer WX₂. A detailed study of the modulation of the valley polarization by interlayer coupling should help unveil the physics behind the unexpectedly large circular polarization in bilayer WS₂ and provide new insights into the future development of valleytronic devices based on TMDs. In this work, we explore the origin of the

anomalously high valley polarization in bilayer WS₂ by a systematic study of samples with different polytypes and on distinct substrates fabricated by different methods. The effect of phonons on valley polarization is clearly revealed by a comparison of valley behaviors in various bilayer WS₂ samples using temperature-dependent circularly polarized PL spectroscopy. Bilayer WS₂ samples whose light emissions are mainly contributed by the direct-gap emission peak possess lowtemperature valley contrast similar to the monolayer counterpart as a result of the suppressed interlayer hopping and the acoustic phonons are shown to be the dominant factor in assisting the intervalley scattering process. And, the depletion of such phonons in the indirect optical transition process could give rise to the exceptionally robust valley polarization in some WS₂ bilayers, where indirect-gap emission contributes significantly to the overall light emission.

Results and discussion

Before we probe the circularly polarized behaviors of bilayer WS₂, we first studied the effects of substrates and fabrication methods on valley contrast in monolayer WS₂ as a reference. Temperature-dependent circularly polarized PL measurements were performed on three kinds of monolayer samples, i.e. chemical vapor deposition (CVD)-grown WS₂ both on sapphire and on SiO₂/Si substrates, and a ME one on a SiO₂/Si substrate. Fig. S1 (ESI[†]) shows the evolution of PL spectra for the three monolayer WS₂ samples from 80 to 300 K. The PL emission of monolayer WS₂ consists of two components corresponding to the direct excitonic transition at the K point in the Brillouin zone: neutral excitons (A) at higher energy and trions (A⁻) at relatively lower energy.^{31,32} Though the relative spectral weights of A and A⁻ may vary for different types of monolayer WS₂ samples, they all share similar trends of A⁻-A spectral weight transformation with increasing temperature, which can be explained by the escape of electrons from the trion state owing to thermal fluctuations.^{33,34} The degree of circular polarization is calculated as²⁶

$$\rho = \frac{I(\sigma_{-}) - I(\sigma_{+})}{I(\sigma_{-}) + I(\sigma_{+})} \tag{1}$$

where $I(\sigma_{-})$ and $I(\sigma_{+})$ represent the intensities of the corresponding circularly polarized luminescence emissions. The valley contrast of the A and A⁻ peaks in CVD-grown WS₂ on the sapphire substrate is slightly larger than that in the other two samples at 80 K (see Fig. S2 of the ESI[†]), which indicates the trivial effect of the substrate and fabrication method on valley contrast in monolayer WS₂.

We then studied the valley behaviors of bilayer WS₂. When two individual layers are vertically stacked, different bilayer structures can be formed depending on the twist angle θ , among which the most stable configurations are an AA stacking sequence when $\theta = 0^{\circ}$ and an AB stacking (also called Bernal stacking) sequence when $\theta = 60^{\circ}$.³⁵ To begin with, we compared the valley behaviors of AA- and AB-stacked bilayer WS₂ on SiO₂/Si substrates (Fig. 1). According to the relative orientation



Fig. 1 Circularly polarized behaviors of CVD-grown AA- and AB-stacked bilayer WS_2 on SiO_2/Si substrates. Circularly polarized components of PL spectra as a function of temperature for (a) AA- and (b) AB-stacked bilayer WS_2 . (c) and (d) show optical images of AA- and AB-stacked bilayer WS_2 samples, respectively. (e) Temperature dependence of circular polarization for AA- and AB-stacked bilayer WS_2 . The excitation light is left-circularly polarized in all the measurements.

of the two vertically stacked layers from the optical images in Fig. 1c and d, the twist angle and the corresponding stacking order can be determined.³⁵ Fig. 1a and b present the circularly polarized components of PL spectra as a function of temperature for AA- and AB-stacked bilayer WS2. The A peak dominates the PL spectra and redshifts with the increase of temperature, which is attributed to the reduction of the direct transition energy caused by the expanded in-plane lattice constant.³⁶ As can be seen in Fig. 1e, both these two samples have similar dependences of valley contrast on temperature and the valley contrast at 80 K is close to that of the monolayer WS₂ on the SiO₂/Si substrate. This observation is due to the suppressed interlayer hopping in these two samples. In AB-stacked bilayer WS₂, it is caused by the large spin-valley coupling of holes compared to the interlayer hopping amplitude,^{25,37} while the opposite spin of electrons between neighbouring layers inhibits the interlayer hopping in AA-stacked bilayer WS2.17,24,38 By increasing the integration time, the indirect bandgap peak (denoted I) corresponding to Λ - Γ transition³⁹ could also be observed, as shown in Fig. S3 of the ESI.† Though the I peak is much broader than the A exciton peak, the integrated intensity of the former is only slightly larger than one quarter of that of the latter, which makes it difficult to resolve the I peak under the measurement conditions in Fig. 1. Since the polarization of the weak I peak in bilayer WS₂ on the SiO₂/Si substrate is not the focus of this paper, we did not use long integration times for the collection of PL spectra at low temperatures to avoid laser-induced damage.

To investigate the effect of the fabrication method on valley contrast in bilayer WS_2 , temperature-dependent circularly polarized PL measurements were performed on ME bilayer WS_2 . Fig. 2 shows the circularly polarized behaviors of ME bilayer WS_2 on a SiO₂/Si substrate as a function of temperature. Since ME bilayer WS_2 was exfoliated from a bulk 2H-WS₂



Fig. 2 Circularly polarized behaviors of ME bilayer WS₂ on a SiO₂/Si substrate. (a) Circularly polarized components of PL spectra of as a function of temperature for ME bilayer WS₂ on a SiO₂/Si substrate. (b) Optical image of the ME bilayer WS₂ sample. (c) Temperature dependence of circular polarization for ME bilayer WS₂. The excitation light is left-circularly polarized in all the measurements.

crystal, it is supposed to possess an AB stacking sequence.⁴⁰ As shown in Fig. 2a, the obvious *I* peak appears over the whole temperature range and exhibits a blueshift with increasing temperature, consistent with a previous report.³⁶ In contrast to the remarkable valley-polarized behaviors of A excitons, the *I* peak is circularly unpolarized at all the measured temperatures, in good agreement with a previous observation.¹⁴ A comparison of Fig. 1e and 2c shows that the valley polarization of A excitons in ME bilayer WS₂ is obviously larger than that of the CVD-grown AB-stacked bilayer WS₂ on the SiO₂/Si substrate and can reach 64.3% at 80 K. Considering that ME monolayer WS₂ shows comparable valley polarization to that of the CVD-grown monolayer WS₂, the pure effect of the fabrication method on the valley polarization of individual layers can be excluded and

intervalley scattering may possibly be weakened by the interlayer coupling in ME bilayer WS₂ on the SiO₂/Si substrate.

To probe whether the underlying substrate could influence the intervalley scattering, we then measured the circularly polarized behaviors of CVD-grown AA-stacked bilayer WS2 on the sapphire substrate (Fig. 3). As shown in Fig. 3a and b, CVD-grown AA-stacked bilayer WS2 on the sapphire substrate can be classified into two types based on their distinct valley polarization and light emission pattern. For the first kind of sample (marked as AA-1) with relatively large valley polarization (see also Fig. 3e), PL spectra contain obvious contribution of the I peak apart from the A exciton peak at high temperatures. In contrast, the I peak is much weaker for the other kind of bilayer sample (marked as AA-2) and its valley polarization is close to that of the CVD-grown AA-stacked bilayer WS₂ on the SiO₂/Si substrate. Therefore, it could be inferred that the valley polarization is associated with the intensity of the I peak. According to the calculated structural energy for WS₂ bilayers with different stacking patterns, there are two possible highsymmetry AA stacking configurations, with the S atoms in the top layer placed above either the W atoms or the S atoms in the bottom layer.²⁹ Due to steric effects, the interlayer distances are different for these two configurations,³⁵ which may result in different interlayer coupling interactions and intensities of the I peak. The varying interlayer coupling strength in different CVD-grown bilayer WS₂ samples will be further discussed later. In addition, the sharp and strong emission from the sapphire substrate appears at around 693 nm⁴¹ in all PL spectra and is circularly unpolarized over the whole temperature range. We also performed the temperature-dependent circularly polarized PL measurement on AB-stacked bilayer WS₂ on a sapphire substrate (see Fig. S4 of the ESI⁺), but the signal-to-noise ratio is too low to resolve the valley behaviors due to the vanished

interference effect, which fails to enhance the efficiency of light absorption and collection unlike a SiO_2/Si substrate.⁴²

Finally, to obtain a comprehensive view of the correlation between the valley contrast and I peak, we compared the unpolarized PL spectra at room temperature and the corresponding valley behaviors of different types of bilayer WS₂ samples in Fig. 4. It can be clearly seen that the valley polarization is indeed closely related to the intensity of the I peak in the bilayer WS₂ samples (see Fig. 4a and b). More specifically, the bilayer WS_2 samples with a relatively strong I peak have a larger valley contrast than those with relatively weak indirect bandgap emission. The link can be explained by phonons which play an important role in both the indirect transition and intervalley scattering process.^{14,35,43} The indirect transition in both AA- and AB-stacked bilayer WS2 occurs between Γ and Λ points^{29,36} with the assistance of Λ -point phonons for the momentum conservations. On the other hand, the intervalley scattering between K and K' points requires the participation of a phonon with the corresponding wave vector difference,⁴³ twice as large as that of the previous one which assists the indirect transition process.¹⁵ Therefore, the existence of a relatively strong I peak decreases the possibility of intervalley scattering by depleting phonons which may have participated in this process in pairs, leading to enhanced valley contrast. Among the three bilayer WS₂ samples with a relatively weak indirect-gap emission peak, AA-stacked bilayer WS₂ on the sapphire substrate has larger valley contrast than those on the SiO₂/Si substrates at low temperature, which is attributed to the inheritance of the valley behaviors of monolayer WS₂ on the two substrates (see Fig. S2 of the ESI[†]). To further confirm the relationship between valley polarization and the I peak, we explored the temperature dependence of valley polarization in five bilayer WS₂ samples. Phonons are bosons and their



Fig. 3 Circularly polarized behaviors of CVD-grown AA-stacked bilayer WS_2 on a sapphire substrate. (a) Circularly polarized components of PL spectra as a function of temperature for AA-stacked bilayer WS_2 with a relatively (a) strong and (b) weak *I* peak (marked as AA-1 and AA-2, respectively). (c) and (d) show optical images of the two bilayer WS_2 samples corresponding to AA-1 and AA-2. (e) Temperature dependence of circular polarization for these two kinds of AA-stacked bilayer WS_2 samples. The excitation light is left-circularly polarized in all the measurements.



Fig. 4 Dependence of circular polarization on the l peak. (a) PL spectra of different types of bilayer WS₂ samples at 300 K. The PL features of sapphire were subtracted from the spectra of bilayer WS₂ on a sapphire substrate for clarity. (b) Temperature dependence of circular polarization for different types of bilayer WS₂ samples.

number at equilibrium obeys Bose–Einstein statistics. As shown in Fig. 4b, for the bilayer samples which do not show a strong Ipeak, their circular polarization dependence on temperature can be well fitted by eqn (2), which is inversely proportional to the phonon population.

$$\rho \propto \left[\exp(E_{\rm p}/k_{\rm B}T) - 1 \right] \tag{2}$$

Table 1 lists the fitted energies and wavenumbers of phonons from fitting, together with the assignment of phonons based on the phonon dispersion of WS2.44 As discussed previously, the phonon mode at the Λ point could work in pair to satisfy the momentum conservation during the intervalley scattering. The three phonon modes at the Λ point with the appropriate energy⁴⁴ further support our explanation of the high valley polarization in bilayer WS_2 with a relatively strong I peak by phonon depletion in the indirect recombination process. Our finding is in agreement with one recent work on the origin of the double-resonance process in TMDs.45 By probing Raman spectra as a function of the laser excitation energy, Carvalho et al. claimed that acoustic phonons dominate the intervalley scattering of electrons and possibly destruct the valley polarization in MoS₂.⁴⁵ On the other hand, for the bilayer samples with a relatively strong I peak, the change in the circular polarization degree versus temperature, especially the values in the low temperature range as marked in Fig. 4b, cannot be fitted well by this expression. This is because phonons are partly depleted in the indirect transition process and the effective number of phonons which can assist the intervalley scattering does not obey Bose-Einstein statistics anymore.

Table 1 The energies, wavenumbers, and assignments of the phonons that participate in the intervalley scattering of electrons for different types of bilayer WS_2 samples

	$E_{\rm p}$ (meV)	$\omega_{\rm p} ({\rm cm^{-1}})$	Assignment
CVD AA on SiO ₂	17.13	138.1	ZA (Λ)
CVD AB on SiO ₂	19.15	154.5	$LA(\Lambda)$
CVD AA on sapphire	12.55	101.2	TA (Λ)
(without any strong <i>I</i> peak)			

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From Fig. 4a, it should also be noticed that both the peak position and width vary for the PL spectra of different types of bilayer WS₂ samples. The redshift of the I peak in AA-stacked bilayer WS₂ compared with ME bilayer WS₂ is due to the lower energy of indirect transition for AA-stacked bilayer WS₂ according to theoretical calculation.²⁹ For the A exciton peak, the relative shift is attributed to the combined effect of interlayer coupling and local strain induced by the sample-substrate interactions. A comparison of PL spectra for the samples on the same substrate (i.e. ME versus CVD-grown AA/AB-stacked bilayer WS2 on SiO2/Si and CVD-grown AA-1 versus AA-2 bilayer WS_2 on sapphire) demonstrates that the energies of the A exciton peak for bilayer WS₂ with a relatively stronger I peak generally redshift compared to those with a weak indirect-gap emission peak. The weak I peak manifests relatively weakened interlayer coupling and therefore the peak positions of the A excitons for bilayer WS2 with a relatively weak indirect-gap emission peak exhibit blueshifts towards those of monolayer WS₂ at higher energies.^{39,46} To further confirm that the shift is indeed caused by interlayer coupling, we compared the frequency differences of the $E^1_{2g}(\Gamma)$ and $A_{1g}(\Gamma)$ modes in different types of bilayer WS2 samples on SiO2/Si substrates, which are indicative of the interlayer coupling strength.^{35,47} As shown in Fig. S5 of the ESI,[†] the separation of the $E_{2g}^1(\Gamma)$ and $A_{1g}(\Gamma)$ modes is larger in ME bilayer WS₂ than those of the CVD-grown bilayer counterparts, which demonstrates stronger interlayer coupling in the ME sample and provides support for our explanation of the exciton shift for samples on the same substrate from the viewpoint of interlayer coupling. The distinct strength of interlayer coupling is attributed to the change in interlayer distances in the different types of bilayer WS2 samples since van der Waals interaction is negatively related to the interlayer distances.^{35,48} In contrast to the ME bilayer WS_2 sample, the uncontrollable variation of the interlayer distance in different types of CVD-grown bilayer samples could lead to varying interlayer coupling strength.³⁵ Future work will investigate the underlying cause of the varying interlayer coupling strength in different CVD-grown bilayer WS₂ samples, even on the same sapphire substrate. Moreover, for the same kind of bilayer WS₂ sample (*i.e.* bilayer WS₂ with either a strong or a

weak I peak) on different substrates, the relative shift of the A exciton peak might be due to the local strain induced by the sample-substrate interactions during the exfoliation or growth process, which has been commonly observed in atomically thin layered materials.^{49–54} The lower surface roughness of sapphire compared to SiO₂⁴⁹ makes the conformally adhered bilayer WS₂ on it subject to smaller tensile strain and leads to higher energies of the exciton peak for bilayer WS₂ on the sapphire substrate.32,55 This is consistent with our observation of the exciton redshift in ME (CVD-grown AA/AB-stacked) bilayer WS₂ on the SiO₂/Si substrate relative to the CVD-grown AA-2 (AA-1) counterpart on the sapphire substrate. Furthermore, it is known that the exciton-phonon interactions broaden the linewidths of excitons.⁵⁶ For the variation of exciton width as shown in Fig. 4a, the ME and CVD-grown AA-1 bilayer WS₂ have broader A exciton peaks than the other bilayer samples, which reflects the stronger exciton-phonon interaction in these two types of samples.

Experimental

Monolayer and bilayer WS₂ samples were fabricated by mechanical exfoliation and grown by the CVD method, followed by temperature-dependent circularly polarized PL measurements.

Sample preparation

CVD monolayer and bilayer WS_2 samples were grown on 300 nm SiO_2/Si and sapphire substrates by a previously reported technique.⁵⁷ Compared with the experimental process to grow monolayer WS_2 , the growth time at 750 °C was increased for bilayer WS_2 , which could make vertical layer-by-layer growth more preferable.³⁵ Exfoliated WS_2 flakes used in this work were obtained by mechanical cleavage of a bulk 2H-WS₂ crystal (2D Semiconductors, CAS #12138-09-9, USA) and then transferred onto 300 nm SiO₂/Si substrates. Monolayer and bilayer WS_2 flakes were selected using an optical microscope (Olympus BX 51), followed by confirmation through PL/Raman spectroscopy.

Optical characterization

The PL measurements were performed with a WITec CRM200 confocal Raman system under a 532 nm excitation laser with a 150 lines per mm grating. To achieve polarization-resolved PL measurements, the linearly polarized light was first transformed into left-circularly polarized light by passing through an achromatic quarter-wave plate (WPQ10M-532, Thorlabs) and then shone on the sample. Next, the emitted backscattering beam comprising both left- and right-circularly light got through the same quarter-wave plate and was separated into two perpendicularly polarized beams, which could be separately detected by rotating the linear polarizer located on top of the edge filter. The accuracy of the circular polarization degree for our setup was no less than 94.0%. The linear polarizer was rotated cautiously to avoid any perturbation effect on the collimation of the whole system throughout the experiment. After both the left- and right-circularly polarized components of PL spectra were measured,

the linear polarizer was rotated back to the original position to confirm that the measured spectrum was fully reproducible. To achieve a fair comparison by ruling out the hysteresis effect, all temperature-dependent PL measurements were performed from room temperature down to 80 K. The cooling of the samples was realized using a temperature controlled stage HFS600E (Linkam Scientific Instruments). At each temperature, at least two different points were tested to confirm that the observed trend of the spectral change was typical. A longworking distance 50× objective lens was used in all the PL measurements with a numerical aperture of 0.55. Raman measurements were carried out by the same WITec system under a 532 nm excitation laser with a 1800 grooves per mm grating at room temperature. The laser spot was evaluated to be around 1 µm. The laser power at the sample surface was maintained with great care to avoid laser-induced damage and kept unchanged during the temperature-dependent measurement.

Conclusions

In summary, we investigated temperature-dependent valley polarization in various AA- and AB-stacked bilayer WS₂ samples and experimentally demonstrated the correlation between the intensity of the indirect bandgap emission and valley polarization. We showed that acoustic phonons play a leading role in the valley polarization of bilayer WS₂ and the depletion of such phonons in the indirect optical transition process could contribute to the exceptionally robust valley polarization. Our results shed light on the critical role of electron–phonon coupling in the intervalley scattering process in bilayer WS₂ and lay a foundation for developing valleytronic applications exploiting the strong electron–phonon interaction in 2D TMDs. Our findings provide new insights into the realization of more robust valley polarization in broader 2D systems by manipulating the indirect optical transition and intervalley scattering.

Conflicts of interest

There are no conflicts to declare.

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References

- 1 S. Jo, N. Ubrig, H. Berger, A. B. Kuzmenko and A. F. Morpurgo, *Nano Lett.*, 2014, **14**, 2019–2025.
- T. Georgiou, R. Jalil, B. D. Belle, L. Britnell, R. V. Gorbachev, S. V. Morozov, Y.-J. Kim, A. Gholinia, S. J. Haigh, O. Makarovsky, L. Eaves, L. A. Ponomarenko, A. K. Geim, K. S. Novoselov and A. Mishchenko, *Nat. Nanotechnol.*, 2013, 8, 100–103.
- 3 R. C. Cooper, C. Lee, C. A. Marianetti, X. Wei, J. Hone and J. W. Kysar, *Phys. Rev. B: Condens. Matter Mater. Phys.*, 2013, 87, 035423.
- 4 Y. Zhang, J. Ye, Y. Matsuhashi and Y. Iwasa, *Nano Lett.*, 2012, **12**, 1136–1140.
- 5 C. Cong, J. Shang, Y. Wang and T. Yu, *Adv. Opt. Mater.*, 2018, 6, 1700767.
- 6 G. B. Liu, D. Xiao, Y. Yao, X. Xu and W. Yao, *Chem. Soc. Rev.*, 2015, **44**, 2643–2663.
- 7 C. Qin, Y. Gao, Z. Qiao, L. Xiao and S. Jia, Adv. Opt. Mater., 2016, 4, 1429–1456.
- 8 H. Kwon, K. Lee, J. Heo, Y. Oh, H. Lee, S. Appalakondaiah,
 W. Ko, H. W. Kim, J. W. Jung, H. Suh, H. Min, I. Jeon,
 E. Hwang and S. Hwang, *Adv. Mater.*, 2017, 29, 1702931.
- 9 F. Barati, M. Grossnickle, S. Su, R. K. Lake, V. Aji and N. M. Gabor, *Nat. Nanotechnol.*, 2017, **12**, 1134–1139.
- 10 H. Zeng and X. Cui, Chem. Soc. Rev., 2015, 44, 2629-2642.
- Y. Wan, J. Xiao, J. Li, X. Fang, K. Zhang, L. Fu, P. Li, Z. Song,
 H. Zhang, Y. Wang, M. Zhao, J. Lu, N. Tang, G. Ran,
 X. Zhang, Y. Ye and L. Dai, *Adv. Mater.*, 2018, **30**, 1703888.
- 12 X. Chen, T. Yan, B. Zhu, S. Yang and X. Cui, ACS Nano, 2017, 11, 1581–1587.
- 13 C. F. Hirjibehedin, Nat. Phys., 2013, 9, 756-757.
- 14 B. Zhu, H. Zeng, J. Dai, Z. Gong and X. Cui, Proc. Natl. Acad. Sci. U. S. A., 2014, 111, 11606–11611.
- 15 K. F. Mak, K. He, J. Shan and T. F. Heinz, *Nat. Nanotechnol.*, 2012, 7, 494–498.
- 16 A. M. Jones, H. Yu, N. J. Ghimire, S. Wu, G. Aivazian, J. S. Ross, B. Zhao, J. Yan, D. G. Mandrus, D. Xiao, W. Yao and X. Xu, *Nat. Nanotechnol.*, 2013, 8, 634–638.
- 17 R. Suzuki, M. Sakano, Y. J. Zhang, R. Akashi, D. Morikawa, A. Harasawa, K. Yaji, K. Kuroda, K. Miyamoto, T. Okuda, K. Ishizaka, R. Arita and Y. Iwasa, *Nat. Nanotechnol.*, 2014, 9, 611–617.
- 18 W. Yao, D. Xiao and Q. Niu, *Phys. Rev. B: Condens. Matter Mater. Phys.*, 2008, 77, 235406.
- 19 H. Yu, X. Cui, X. Xu and W. Yao, Natl. Sci. Rev., 2015, 2, 57-70.
- 20 D. Xiao, G. B. Liu, W. Feng, X. Xu and W. Yao, *Phys. Rev. Lett.*, 2012, **108**, 196802.
- 21 D. Lagarde, L. Bouet, X. Marie, C. R. Zhu, B. L. Liu, T. Amand, P. H. Tan and B. Urbaszek, *Phys. Rev. Lett.*, 2014, **112**, 047401.
- 22 G. Kioseoglou, A. T. Hanbicki, M. Currie, A. L. Friedman and B. T. Jonker, *Sci. Rep.*, 2016, **6**, 25041.
- 23 S. Wu, J. S. Ross, G.-B. Liu, G. Aivazian, A. Jones, Z. Fei,
 W. Zhu, D. Xiao, W. Yao, D. Cobden and X. Xu, *Nat. Phys.*,
 2013, 9, 149–153.

- 24 T. Jiang, H. Liu, D. Huang, S. Zhang, Y. Li, X. Gong, Y. R. Shen, W. T. Liu and S. Wu, *Nat. Nanotechnol.*, 2014, 9, 825–829.
- 25 Z. Gong, G. B. Liu, H. Yu, D. Xiao, X. Cui, X. Xu and W. Yao, *Nat. Commun.*, 2013, 4, 2053.
- 26 A. M. Jones, H. Yu, J. S. Ross, P. Klement, N. J. Ghimire, J. Yan, D. G. Mandrus, W. Yao and X. Xu, *Nat. Phys.*, 2014, 10, 130–134.
- 27 P. K. Nayak, F. C. Lin, C. H. Yeh, J. S. Huang and P. W. Chiu, *Nanoscale*, 2016, **8**, 6035–6042.
- 28 Q. Liu, X. Zhang and A. Zunger, Phys. Rev. Lett., 2015, 114, 087402.
- 29 J. He, K. Hummer and C. Franchini, *Phys. Rev. B: Condens. Matter Mater. Phys.*, 2014, **89**, 075409.
- 30 S. Zheng, L. Sun, X. Zhou, F. Liu, Z. Liu, Z. Shen and H. J. Fan, *Adv. Opt. Mater.*, 2015, 3, 1600–1605.
- 31 G. Plechinger, P. Nagler, J. Kraus, N. Paradiso, C. Strunk, C. Schüller and T. Korn, *Phys. Status Solidi RRL*, 2015, 9, 457–461.
- 32 Y. Wang, C. Cong, W. Yang, J. Shang, N. Peimyoo, Y. Chen, J. Kang, J. Wang, W. Huang and T. Yu, *Nano Res.*, 2015, 8, 2562–2572.
- 33 J. S. Ross, S. Wu, H. Yu, N. J. Ghimire, A. M. Jones, G. Aivazian, J. Yan, D. G. Mandrus, D. Xiao, W. Yao and X. Xu, *Nat. Commun.*, 2013, 4, 1474.
- 34 X. X. Zhang, Y. You, S. Y. Zhao and T. F. Heinz, *Phys. Rev. Lett.*, 2015, **115**, 257403.
- 35 K. Liu, L. Zhang, T. Cao, C. Jin, D. Qiu, Q. Zhou, A. Zettl, P. Yang, S. G. Louie and F. Wang, *Nat. Commun.*, 2014, 5, 4966.
- 36 W. Zhao, R. M. Ribeiro, M. Toh, A. Carvalho, C. Kloc, A. H. Castro Neto and G. Eda, *Nano Lett.*, 2013, 13, 5627–5634.
- 37 H. Zeng, G. B. Liu, J. Dai, Y. Yan, B. Zhu, R. He, L. Xie, S. Xu,
 X. Chen, W. Yao and X. Cui, *Sci. Rep.*, 2013, 3, 1608.
- 38 R. Akashi, M. Ochi, S. Bordács, R. Suzuki, Y. Tokura, Y. Iwasa and R. Arita, *Phys. Rev. A: At., Mol., Opt. Phys.*, 2015, 4, 014002.
- 39 W. Zhao, Z. Ghorannevis, L. Chu, M. Toh, C. Kloc, P.-H. Tan and G. Eda, ACS Nano, 2012, 7, 791–797.
- 40 S. W. Han, H. Kwon, S. K. Kim, S. Ryu, W. S. Yun, D. H. Kim, J. H. Hwang, J. S. Kang, J. Baik, H. J. Shin and S. C. Hong, *Phys. Rev. B: Condens. Matter Mater. Phys.*, 2011, 84, 045409.
- 41 V. Kudryashov, S. Mamakin and A. Yunovich, *Tech. Phys. Lett.*, 1999, **25**, 536–537.
- 42 M. Buscema, G. A. Steele, H. S. J. van der Zant and A. Castellanos-Gomez, *Nano Res.*, 2015, 7, 561–571.
- 43 H. Zeng, J. Dai, W. Yao, D. Xiao and X. Cui, *Nat. Nanotechnol.*, 2012, 7, 490–493.
- 44 A. Molina-Sánchez and L. Wirtz, *Phys. Rev. B: Condens. Matter Mater. Phys.*, 2011, **84**, 155413.
- 45 B. R. Carvalho, Y. Wang, S. Mignuzzi, D. Roy, M. Terrones, C. Fantini, V. H. Crespi, L. M. Malard and M. A. Pimenta, *Nat. Commun.*, 2017, 8, 14670.
- 46 H. R. Gutierrez, N. Perea-Lopez, A. L. Elias, A. Berkdemir, B. Wang, R. Lv, F. Lopez-Urias, V. H. Crespi, H. Terrones and M. Terrones, *Nano Lett.*, 2013, 13, 3447–3454.

- 47 A. M. van der Zande, J. Kunstmann, A. Chernikov, D. A. Chenet, Y. You, X. Zhang, P. Y. Huang, T. C. Berkelbach, L. Wang, F. Zhang, M. S. Hybertsen, D. A. Muller, D. R. Reichman, T. F. Heinz and J. C. Hone, *Nano Lett.*, 2014, 14, 3869–3875.
- 48 S. Huang, X. Ling, L. Liang, J. Kong, H. Terrones, V. Meunier and M. S. Dresselhaus, *Nano Lett.*, 2014, **14**, 5500–5508.
- 49 H. Sahin, S. Tongay, S. Horzum, W. Fan, J. Zhou, J. Li, J. Wu and F. M. Peeters, *Phys. Rev. B: Condens. Matter Mater. Phys.*, 2013, 87, 165409.
- 50 B. G. Shin, G. H. Han, S. J. Yun, H. M. Oh, J. J. Bae, Y. J. Song, C. Y. Park and Y. H. Lee, *Adv. Mater.*, 2016, 28, 9378–9384.
- 51 Z. Liu, M. Amani, S. Najmaei, Q. Xu, X. Zou, W. Zhou, T. Yu, C. Qiu, A. G. Birdwell, F. J. Crowne, R. Vajtai, B. I. Yakobson, Z. Xia, M. Dubey, P. M. Ajayan and J. Lou, *Nat. Commun.*, 2014, 5, 5246.

- 52 Z.-Q. Xu, Y. Zhang, S. Lin, C. Zheng, Y. L. Zhong, X. Xia, Z. Li, P. J. Sophia, M. S. Fuhrer and Y.-B. Cheng, *ACS Nano*, 2015, 9, 6178–6187.
- 53 R. V. Gorbachev, I. Riaz, R. R. Nair, R. Jalil, L. Britnell,
 B. D. Belle, E. W. Hill, K. S. Novoselov, K. Watanabe,
 T. Taniguchi, A. K. Geim and P. Blake, *Small*, 2011, 7, 465–468.
- 54 W. G. Cullen, M. Yamamoto, K. M. Burson, J. H. Chen, C. Jang, L. Li, M. S. Fuhrer and E. D. Williams, *Phys. Rev. Lett.*, 2010, **105**, 215504.
- 55 A. Kumar and P. K. Ahluwalia, *Modell. Simul. Mater. Sci.* Eng., 2013, **21**, 065015.
- 56 S. Rudin, T. Reinecke and B. Segall, *Phys. Rev. B: Condens. Matter Mater. Phys.*, 1990, **42**, 11218–11231.
- 57 C. Cong, J. Shang, X. Wu, B. Cao, N. Peimyoo, C. Qiu, L. Sun and T. Yu, *Adv. Opt. Mater.*, 2014, **2**, 131–136.